

# A New Molybdenum Nitride Catalyst with Rhombohedral MoS<sub>2</sub> Structure for Hydrogenation Applications

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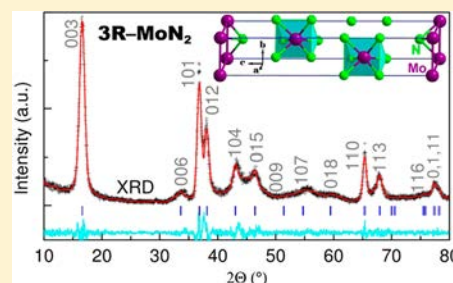
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## Supporting Information

**ABSTRACT:** Nitrogen-rich transition-metal nitrides hold great promise to be the next-generation catalysts for clean and renewable energy applications. However, incorporation of nitrogen into the crystalline lattices of transition metals is thermodynamically unfavorable at atmospheric pressure; most of the known transition metal nitrides are nitrogen-deficient with molar ratios of N:metal less than a unity. In this work, we have formulated a high-pressure route for the synthesis of a nitrogen-rich molybdenum nitride through a solid-state ion-exchange reaction. The newly discovered nitride, 3R-MoN<sub>2</sub>, adopts a rhombohedral R3m structure, isotypic with MoS<sub>2</sub>. This new nitride exhibits catalytic activities that are three times more active than the traditional catalyst MoS<sub>2</sub> for the hydrodesulfurization of dibenzothiophene and more than twice as high in the selectivity to hydrogenation. The nitride is also catalytically active in sour methanation of syngas with >80% CO and H<sub>2</sub> conversion at 723 K. Our formulated route for the synthesis of 3R-MoN<sub>2</sub> is at a moderate pressure of 3.5 GPa and, thus, is feasible for industrial-scale catalyst production.



## INTRODUCTION

Heterogeneous catalysis plays a central role in key chemical processes associated with petrochemical and renewable energy generation. Since the first success in catalytic liquefaction of coal in 1920s,<sup>1</sup> MoS<sub>2</sub>-based catalysts have been extensively studied and used for hydrotreating of fossil fuels,<sup>1</sup> sour methanation of syngas,<sup>2</sup> electrocatalysis for water splitting,<sup>3</sup> and solar energy conversion.<sup>4</sup> The excellent catalytic performances of this family of catalysts are primarily attributed to the layered structure of MoS<sub>2</sub> in which the interlayers of Mo and S are held together by weak van der Waals forces.<sup>5,6</sup> This structural characteristic allows producing a variety of nano-clusters with highly exposed active edge sites on the S–Mo–S sandwich layer. The catalytic activity of the exfoliated MoS<sub>2</sub> thin layers,<sup>7</sup> for example, is three times that of the dispersed MoS<sub>2</sub> on a silica mesoporous substrate.<sup>8</sup> In addition, the edge sites of the S–Mo–S layer can absorb promoter species, such as Co and Ni, significantly improving the catalytic performances.<sup>1,8</sup>

Despite several decades of searches for other catalysts, the transition metal (TM) disulfides remain the most studied and used catalysts in deep hydroprocessing of fossil fuels including hydrodesulfurization (HDS) and hydrodenitrogenation (HDN).<sup>1,9</sup> Over the past decade, however, TM nitrides have attracted considerable attention because they exhibit a number of unique and advanced catalytic properties for hydrotreating (e.g., Mo<sub>2</sub>N and W<sub>2</sub>N)<sup>10</sup> and photo- and electrochemical-catalysis (e.g., δ-MoN, Mo<sub>2</sub>N, and Ta<sub>3</sub>N<sub>5</sub>),<sup>11–13</sup> although none of them adopts a layered structure. One important property is their excellent resistance to corrosion by sulfur, a common catalyst poison;<sup>1,2</sup> hence, they can be used for sour catalysis applications. In some reactions such as dehydrogenation and hydrogenolysis, molybdenum nitride shows similar catalytic activities as platinum-group catalysts.<sup>10,12,14,15</sup> Thus, nitrogen-rich TM nitrides, particularly those of MoS<sub>2</sub>-type, may hold

Received: February 11, 2015

Published: March 23, 2015

great promise as the next-generation catalysts for a wide range of applications.

Preparation of nitride-rich TM nitrides is challenging because incorporation of nitrogen into the crystal lattice of TMs is often thermodynamically unfavorable at atmospheric pressure.<sup>16</sup> As a result, most of the known TM nitrides are nitrogen-deficient with  $x \leq 1$  in  $\text{TMN}_x$ , in spite of the fact that TMs often exhibit oxidation states exceeding +3. To date, reports on TM nitrides with  $x > 1$  are rather fragmentary, sometimes even contradictory, especially on their crystal structures. In the binary Mo–N system, three different phases with varying nitrogen concentrations have been reported: tetragonal  $\beta$ - $\text{MoN}_x$  with  $x \leq 0.5$  (e.g.,  $\text{Mo}_2\text{N}^{17}$ ), cubic  $\gamma$ - $\text{MoN}_x$  with  $0.5 \leq x < 1$  (e.g.,  $\text{Mo}_3\text{N}_2^{18}$ ), and hexagonal  $\delta$ - $\text{MoN}_x$  with  $x \geq 1$  (e.g.,  $\text{MoN}^{19}$ ). Also noted is that  $\text{Mo}_3\text{N}_6$  is the only nitride with  $x > 1$  in the binary system Mo–N.<sup>19</sup>

Using high-pressure techniques, a number of novel nitrogen-rich TM nitrides have recently been synthesized<sup>20</sup> including new polymorphs of  $\text{Zr}_3\text{N}_4$ ,<sup>21</sup>  $\text{Hf}_3\text{N}_4$ ,<sup>21,22</sup>  $\text{Ta}_2\text{N}_3$ ,<sup>23</sup> and  $\text{Ta}_3\text{N}_5$ ,<sup>24</sup> and noble metal dinitrides ( $\text{OsN}_2$ ,  $\text{IrN}_2$ , and  $\text{PtN}_2$ ).<sup>25,26</sup> Successful high- $P$  synthesis of nitrides with higher oxidation states demonstrates that pressure can effectively promote the role of  $d$ -electrons in chemical bonding with nitrogen. However, the required pressures in most of the previous synthetic routes are relatively high, in a range of 11–50 GPa, which is beyond the current technological capability for massive, industrial-scale production. Hexagonal  $\text{ReN}_2$  has recently been synthesized through a solid-state reaction at 7.7 GPa and is the only nitride possessing a layered  $\text{MoS}_2$ -type structure.<sup>27</sup> To date, no nitrogen-rich nitride has been synthesized in the Mo–N system, although an oxidation state as high as +6 occurs in other chemical systems (e.g.,  $\text{MoO}_3$ ).

Most recently, we successfully synthesized stoichiometric CrN and a series of novel nitrogen-rich tungsten nitrides (e.g.,  $\text{W}_2\text{N}_3$  and  $\text{W}_3\text{N}_4$ ) through a newly formulated solid-state ion-exchange reactions between  $\text{Na}_2\text{XO}_4$  ( $X = \text{Cr}, \text{W}$ ) and  $h\text{BN}$  at pressures up to 5 GPa.<sup>28,29</sup> In this work, we have extended this high- $P$  methodology to the Mo–N system with a focus on nitrogen-rich phases, leading to the discovery of a novel nitride,  $\text{MoN}_2$ . Here, we report the high- $P$  synthesis, structural characterization, and catalytic property measurements of this novel nitride compound.

## EXPERIMENTAL SECTION

**Starting Materials, Synthesis, and Purification.** High-purity  $\text{Na}_2\text{MoO}_4$  (>99.5%,  $\sim 50 \mu\text{m}$ ) and  $h\text{BN}$  (>99.9%,  $\sim 50 \mu\text{m}$ ) powders in the molar ratio  $\text{Na}_2\text{MoO}_4:\text{BN} = 1:2$  were homogeneously mixed for the synthesis of molybdenum nitrides.  $\text{Na}_2\text{MoO}_4$  was obtained by dehydration of commercially available  $\text{Na}_2\text{MoO}_4 \cdot 2\text{H}_2\text{O}$  in a muffle furnace at 473 K for 12 h.

High  $P$ - $T$  synthetic experiments were carried out using a  $DS 6 \times 14$  MN cubic press at Sichuan University, China.<sup>30</sup> In each experiment, the powder mixture was compacted into a cylindrical pellet of 12 mm in diameter and 10 mm in height, which was then sealed within a molybdenum capsule to prevent possible contamination. The detailed experimental procedures have been described elsewhere.<sup>30</sup> The run products were washed with distilled water to remove the byproduct  $\text{NaBO}_2$  and unreacted  $\text{Na}_2\text{MoO}_4$ , followed by drying in an oven at 348 K. To obtain a nitrogen-rich nitride, the synthesis was performed at a relatively low temperature of 773 K to avoid potential high- $T$  degassing. For each experimental run, the reaction was incomplete even after a prolonged heating of 8–10 h, and the insoluble  $h\text{BN}$  impurity was identified in the purified sample (see Supporting Information Figure S1). Therefore, a two-step reaction route was adopted to eliminate  $h\text{BN}$ ; the purified run products from the first-step

reaction, the nitride and unreacted  $h\text{BN}$ , were mixed with additional  $\text{Na}_2\text{MoO}_4$  in a weight ratio of  $\sim 1:2$  for the second-step synthesis.

**Characterization.** The final products were characterized by X-ray diffraction (XRD) with  $\text{Cu K}\alpha$  radiation and time-of-flight neutron powder diffraction (NPD) performed at the beamline of Nanoscale-Ordered Materials Diffractometer (NOMAD) of SNS, Oak Ridge National Laboratory.<sup>31</sup> The crystal structure was determined from Rietveld analysis of the X-ray and neutron diffraction data using the GSAS software.<sup>32</sup>

Thermogravimetric mass spectrometer (TG–MS) measurements were carried out in an Ar atmosphere on a Setaram TGA-92 instrument equipped with an OmniStar-200 quadrupole mass spectrometer, which was used to qualitatively monitor the outlet gases at high temperatures. In air, a simultaneous TG-DTG (derivative thermogravimetry) measurement was performed to study the oxidation of the new nitride. For each experimental run, phase-pure nitride powders ( $\sim 10 \text{ mg}$ ) were loaded, and automatically heated to a target temperature of 1073 K at a rate of  $10 \text{ K}\cdot\text{min}^{-1}$  and soaked for 30 min before quenching the sample to room temperature. For the TG–MS measurement, the mass-to-charge ratios,  $m/z$ , were qualitatively recorded to monitor  $\text{N}_2$  and nitrogen oxide gases, such as  $\text{N}_2\text{O}$ ,  $\text{NO}$ , and  $\text{NO}_2$ , for possible oxidation. The nitrogen content,  $x$ , in the obtained  $\text{MoN}_x$  was determined by the TGA–DTG via oxidation of  $\text{MoN}_x$ .

Microstructures of the purified sample were characterized by a field emission scanning electron microscopy (SEM) and high-resolution transmission electron microscopy (HRTEM). Energy-dispersive X-ray (EDX) was employed for compositional analyses and the selected area electron diffraction (SAED) was carried out to confirm the crystal structure. X-ray photoelectron spectroscopy (XPS) measurements were performed to determine the electronic structure of the new phase. Using the Archimedes method, the density of the sample was measured on a bulk sample sintered at 3.5 GPa and 753 K for 8 h.

The specific surface area and pore structures of the nitride catalyst were determined from physisorption of  $\text{N}_2$  by using the BET and BJH methods at 77 K and low relative pressure ( $P/P_0$ ) of 0–0.3.<sup>33,34</sup> The measurement was performed in a Tristar-3000 instrument. Traditional  $\text{MoS}_2$  catalyst was also measured for comparison. Before measurement, the two catalysts were treated at 473 K in a vacuum for 6–8 h to remove possible impurity gases adsorbed from the air.

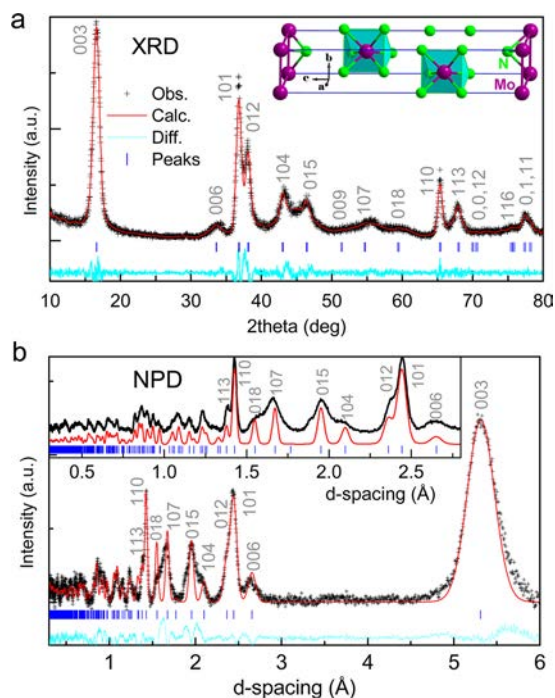
The temperature-programmed desorption (TPD) measurement of the pre-adsorbed CO gas was used to determine the concentration of active Mo atoms in the catalysts. Experiment was conducted in a TP-5080 quartz microreactor. To remove the possible adsorbed impurity gases, the samples ( $\sim 100 \text{ mg}$ ) were treated at 623 K for 30 min in a He atmosphere with a flow rate of  $30 \text{ mL}\cdot\text{min}^{-1}$ , followed by cooling the sample to 323 K over a period of 60 min. Before desorption, chemisorption of CO was carried out at 323 K in a He atmosphere with a 0.4% volume fraction of CO. The mixed gases were flowed for 60 min at a flow rate of  $50 \text{ mL}\cdot\text{min}^{-1}$  to have a sufficient adsorption. In order to eliminate the physisorbed CO, the catalyst was then flushed in a He flow at 323 K for 60 min with a flow rate of  $30 \text{ mL}\cdot\text{min}^{-1}$ . The subsequent desorption process of CO was monitored at temperatures up to 673 K with a heating rate of  $10 \text{ K}\cdot\text{min}^{-1}$ . In all these processes, the volume of the desorbed CO was determined by a quadrupole mass spectrometer, OmniStar GSD-320, which was pre-calibrated by using standard mixed gases of He and CO (see ref 35 for details) and the calibrated parameter is  $\sim 13.147 \text{ mol of CO per unit area}$ .

The catalytic hydrodesulfurization (HDS) of dibenzothiophene (DBT) using  $\text{MoN}_2$  and  $\text{MoS}_2$  catalysts was carried out in a high- $P$  batch reactor/autoclave at 623 K and 6 MPa in  $\text{H}_2$  atmosphere. Before the experiment, DBT was dissolved into decalin as the feedstock in a weight fraction of 500 ppm and the total feed oil is 90.0 g with a volume of 101.1 mL. For each experimental run, 0.3 g of catalyst was placed in the reactor with a stirring speed of 300 rpm. At 623 K, the reaction was followed for 48 h by sampling the products every 8 h. Sour methanation of the syngas was performed in a continuous-flow and fixed-bed reactor. The nitride sample used in the experiment was 0.25 g, which was sandwiched between quartz fibers in the middle of the reactor. A syngas mixture of  $\text{H}_2$ ,  $\text{CO}$ ,  $\text{H}_2\text{S}$ , and  $\text{N}_2$  was used as the

starting reactants in a molar ratio of  $\text{H}_2:\text{CO}:\text{H}_2\text{S}:\text{N}_2 = 47.3:47.3:1.2:5.2$ . The experiment was conducted at 3 MPa and 723 K with an hourly space velocity of  $5000 \text{ h}^{-1}$ . For both HDS and methanation experiments, the outlet gases were quantitatively analyzed by an online gas chromatograph (see refs 36 and 37 for details).

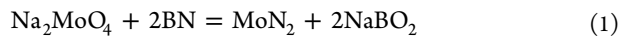
## RESULTS AND DISCUSSION

Figure 1a shows an XRD pattern of the purified product synthesized at 3.5 GPa and 753 K for 20 h. As discussed below,



**Figure 1.** Refined XRD and NPD patterns of 3R-MoN<sub>2</sub>. (a) XRD data collected using a Cu K $\alpha$  radiation. Inset is a polyhedral view of the 3R-MoN<sub>2</sub> structure. (b) NPD data collected at a Bragg scattering angle of 30°. A higher-resolution NPD pattern is plotted in the inset, which was collected at a 66° scattering angle. The observed and calculated patterns are shown in black and red, respectively. The positions of allowed Bragg reflections are denoted by blue tick marks. The sample synthesis was performed at 3.5 GPa and 753 K for 20 h through a two-step reaction (see Experimental Section). The misfit in the  $d$ -spacing range of 1.5–2.1 Å is mainly attributed to significant peak broadening in a subset of reflections, notably 104, 015, 107 and 018, which originates from the layer stacking disorder (refs 5 and 49).

MoN<sub>2</sub> was formed through the following high- $P$  solid-state reaction



which may simply be viewed as an ion-exchange process between Mo<sup>6+</sup> and B<sup>3+</sup>. We have also prepared W<sub>2</sub>N<sub>3</sub> and CrN at high  $P$  and  $T$  conditions through similar reactions.<sup>28,29,38</sup> In all the experimental runs performed, reaction 1 was incomplete even after a prolonged heating of 8–10 h (see Supporting Information Figure S1), indicating that the reaction is kinetically sluggish. The nitride sample can be purified by washing the unreacted Na<sub>2</sub>MoO<sub>4</sub> and reaction by-product, NaBO<sub>2</sub>, with water, leaving behind *h*BN as the only impurity phase in the run product. The unwanted *h*BN can further be eliminated in a two-step reaction route, described in detail in the Experimental Section, leading to a phase-pure nitride after a second washing. The obtained run product displays a number

of Bragg reflections that do not belong to any of the known molybdenum nitrides such as Mo<sub>2</sub>N, Mo<sub>3</sub>N<sub>2</sub>, MoN, Mo<sub>5</sub>N<sub>6</sub>, and their solid solutions (Figure 1a).<sup>17,19,39</sup> Instead, the diffraction peaks can be indexed by a rhombohedral structure with the lattice parameters  $a = 2.854(1) \text{ \AA}$  and  $c = 15.938(2) \text{ \AA}$ . Further, the XRD pattern of this new nitride resembles those of WN<sub>2</sub>,<sup>40</sup> 3R-MoS<sub>2</sub>,<sup>5</sup> and LiXN<sub>2</sub> ( $X = \text{W}$  and  $\text{Mo}$ )<sup>41,42</sup> with a symmetry of either  $R\bar{3}m$  (No. 166) or  $R3m$  (No. 160). In most molybdenum and tungsten nitrides, trigonal-prismatic coordination [XN<sub>6</sub>] ( $X = \text{Mo}$  and  $\text{W}$ ) is preferably adopted for  $X$  atoms.<sup>38,41,42</sup> On the other hand, the  $R\bar{3}m$  structure only contains the octahedral coordination [XN<sub>6</sub>]. Therefore, the space group  $R3m$  (No. 160), in which  $X$  atoms reside in the trigonal prismatic sites, was used for Rietveld analysis of the XRD data.

In the  $R3m$  structure, three crystallographically distinct positions are allowed for Mo with multiplicities of 3, 9, and 18.<sup>43</sup> For most reported binary TM–N compounds, as a constraint, their densities typically cannot exceed those of their parent metals. Thus, the Wyckoff site 3a (0, 0,  $z$ ), which has the lowest number of multiplicity, is the only possible position for Mo in this structure. Because the XRD intensities of molybdenum nitride primarily originate from X-ray scattering of Mo, N was omitted from the structure model during the initial Rietveld analysis. The thus-refined position for Mo is 3a (0, 0,  $z$ ) with  $z = 0$ . Since further treatment of the as-purified nitride at 3.5 GPa and 1523 K led to its decomposition into a known hexagonal  $\delta$ -MoN (see Supporting Information Figure S6), the new nitride phase must be nitrogen-rich with  $y: x > 1$  in Mo <sub>$x$</sub> N <sub>$y$</sub> . However, as an upper limit, the  $x:y$  ratio should not be greater than 2 because the highest possible oxidation state for Mo is +6. In addition, two independent 3a (0, 0,  $z$ ) sites are allowed for N to achieve the experimental density of  $\rho_0 \approx 4.97 \text{ g}\cdot\text{cm}^{-3}$ , determined using a well-sintered bulk sample (see Supporting Information Figure S8 and Table S2). On the basis of these considerations, the stoichiometry of this new nitride compound is determined to be Mo<sub>3</sub>N<sub>6</sub> (i.e., MoN<sub>2</sub>). This formula is in good agreement with the nitrogen concentration obtained from thermogravimetric measurement,  $x \approx 2.2(3)$  in MoN <sub>$x$</sub> , which will be discussed later.

To further resolve the crystal structure, we conducted powder neutron diffraction experiment on the purified run product. Unlike X-rays, the N atom is an excellent neutron scatterer with a coherent scattering length of  $\sim 9.36 \text{ fm}$ , which is  $\sim 40\%$  larger than that of Mo ( $\sim 6.72 \text{ fm}$ ).<sup>44</sup> Therefore, the atomic positions of nitrogen can be well determined from Rietveld analysis of neutron diffraction data (Figure 1b). The thus-obtained coordinates for nitrogen are 3a (0, 0,  $z$ ) with  $z = 0.258(1)$  and  $z = 0.402(2)$ , indicating that Mo ions are in trigonal-prismatic coordination, [MoN<sub>6</sub>] (see inset of Figure 1). The refined structural parameters of MoN<sub>2</sub> are summarized in Table 1. As shown in Figure 1, the calculated X-ray and neutron patterns using this structure model agrees well with the observed patterns. Because the resolved structure for MoN<sub>2</sub> is isotypic with that of rhombohedral 3R-MoS<sub>2</sub>,<sup>5</sup> we denote it as 3R-MoN<sub>2</sub>. Evidently, a subset of diffraction peaks in Figure 1, such as 104, 015, 107, and 018, exhibit significant broadening, whereas other peaks (such as 003, 101, and 110) remain sharp (see Supporting Information Figure S3). In addition, some reflections in the low  $d$ -spacing (i.e., large  $2\theta$ ) region either are too weak to be seen (e.g., 009 and 116) or show significant reduction in the diffraction intensity (e.g., 107 and 018). These phenomena, frequently occurred in layer-structured systems in



**Table 1. Summary of Structural Parameters for MoN<sub>2</sub> Phases Determined by Rietveld Analyses of X-ray and Neutron Diffraction Data Taken at Ambient Conditions**

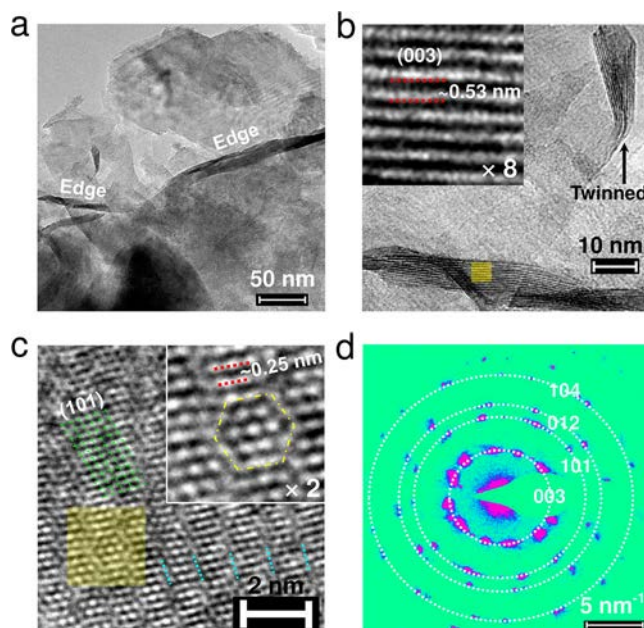
3R-MoN <sub>2</sub>	
formula	Mo <sub>3</sub> N <sub>6</sub>
space group	rhombohedral, <i>R3m</i> (No. 160)
cell parameters [Å]	<i>a</i> = <i>b</i> = 2.854 (1), <i>c</i> = 15.938 (2)
cell volume [Å <sup>3</sup> ]	112.426
density [g·cm <sup>-3</sup> ]	5.492
Mo Wyckoff site	Mo1:3 <i>a</i> , (0, 0, 0.000 (1))
N Wyckoff sites	N1:3 <i>a</i> , (0, 0, 0.258 (1)), N2:3 <i>a</i> , (0, 0, 0.402 (2))
<i>d</i> <sub>Mo-N</sub> [Å]	1.976, 2.037
<i>D</i> <sub>interlayer</sub> [Å] <sup>a</sup>	5.313 (1)
<i>D</i> <sub>layer distance</sub> [Å] <sup>a</sup>	3.017 (1)
<i>R</i> <sub>p</sub> , <i>wR</i> <sub>p</sub> [%] (NPD)	3.2, 5.3

<sup>a</sup>*D*<sub>interlayer</sub> is the distance between the two nearest neighboring Mo planes, and *D*<sub>layer distance</sub> corresponds to the distance between the N planes.

nanocrystalline forms such as nanocrystalline MoS<sub>2</sub>,<sup>45–48</sup> can mainly be attributed to the layer stacking disorder and atomic disorder on the surface of nanocrystals.<sup>5,45,49</sup> As will be discussed below, as-prepared 3R-MoN<sub>2</sub> is also in a nanocrystalline form with layer stacking disorder. It is worthwhile to mention that 3R-MoN<sub>2</sub> and MoS<sub>2</sub> have almost the same core-level Mo 3d and Mo 3p binding energy (see X-ray photoelectron spectroscopy measurement in Supporting Information Figures S4 and S5). The determined valence state for Mo is +3.5 in both 3R-MoN<sub>2</sub> and MoS<sub>2</sub>, inferring a peculiar 4d<sup>2.5</sup> electronic structure. Note that a non-integral valence state (< +4) has previously been reported in MoS<sub>2</sub>.<sup>50</sup> Further computational work is warranted to quantify and understand the interesting electronic structures in 3R-MoN<sub>2</sub>.

The microstructures obtained from high-resolution transmission electron microscopy (HRTEM) are shown in Figure 2. As expected for a layered compound, the grains of 3R-MoN<sub>2</sub> exhibit a laminated morphology with a crystallite size of ~100–200 nm and an edge thickness of ~10–25 nm (see Figure 2a). Figure 2b shows the atomic-scale crystalline edge. Along the *c* axis, the Mo atoms show a fringe pattern with a distance of ~0.53 nm, which corresponds to the (003) plane (also see Table 1). The crystalline edge is also nano-twinned and nano-twisted, corresponding to layer stacking disorder of the N–Mo–N interlayers. As shown in Figure 2c, (101) lattice fringes with a *d*-spacing of ~0.25 nm are clearly revealed; this *d*-spacing is close to the refined value of 2.4716 Å (i.e., *a*√3/2, see Table 1). Consistent with the resolved structure for 3R-MoN<sub>2</sub>, the HRTEM image (inset in Figure 2c) shows a clear hexagon formed by Mo atoms. Also noted is that the crystals along (101) plane show a ribbonlike morphology with distinct grain boundaries. Figure 2d shows a selected-area electron diffraction pattern; four diffraction rings are identified, and they correspond to the planes (003), (101), (012), and (104).

Thermal stabilities of the new nitride were studied by simultaneous thermogravimetry–derivative thermogravimetry (TG–DTG), as shown in Figure 3. In air, there is an obvious weight increase at ~610 K, indicating the onset of oxidation, which is comparable to that of MoS<sub>2</sub> (~623 K). Evidently, the oxidation process is complete at ~780 K when a plateau value is reached, and the final product is MoO<sub>3</sub> (see Supporting Information Figure S7). The nitrogen concentration in the nitride can be readily determined from thermogravimetric



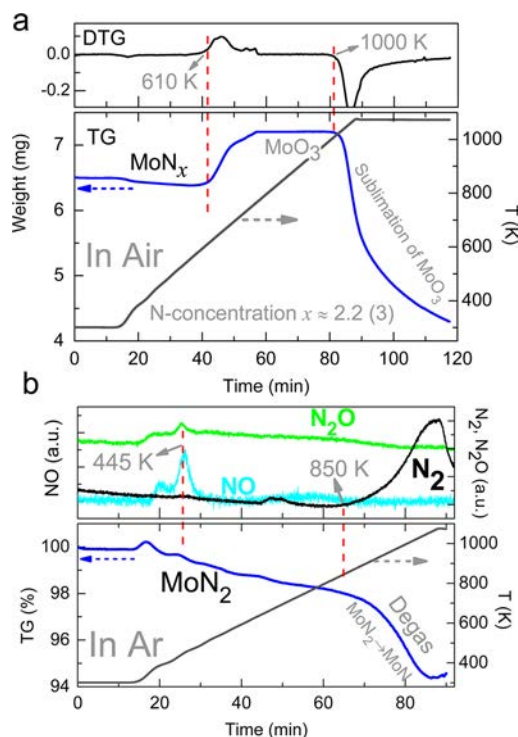
**Figure 2.** HRTEM images and electron diffraction pattern for 3R-MoN<sub>2</sub>. (a) Plate-shaped crystals with laminate edges. (b) Atomic-scale images of the N–Mo–N interlayers, that is, the (003) plane, and (c) the (101) plane. The inset in both (b) and (c) are the enlarged portions of the regions highlighted in yellow. Dotted double red lines indicate the *d*-spacing for the fingerprints of (003) and (101) planes. Dotted cyan lines in (c) denote the grain boundaries. (d) Selected area electron diffraction, SAED. The observed four diffraction rings correspond to the lattice planes of (003), (101), (012), and (104).

data,<sup>38</sup> and the obtained value is *x* = 2.2(3) in MoN<sub>*x*</sub>. The sublimation of MoO<sub>3</sub> happens at ~1000 K as the weight decreases steeply. As shown in Figure 3b, in an Ar atmosphere, the nitrogen degassing of 3R-MoN<sub>2</sub> starts at ~850 K, which gives rise to a more stable product of hexagonal δ-MoN (see Supporting Information Figure S6)<sup>51</sup> via the reaction



The desorption of NO and N<sub>2</sub>O occurs in the temperature region of ~400–450 K, which results from the reaction between MoN<sub>2</sub> and O<sub>2</sub> adsorbed on the surface. This indicates that the NO chemisorption analysis, a common method used for the determination of coordinately unsaturated Mo sites of MoS<sub>2</sub>-based catalysts,<sup>52</sup> cannot be used for 3R-MoN<sub>2</sub>, which is discussed below in more detail.

The catalytic properties of 3R-MoN<sub>2</sub> were measured in hydrodesulfurization (HDS) of dibenzothiophene (DBT), a common sulfur-bearing compound in diesel fuels that is more difficult to be hydrodesulfurized compared with thiols and thiophenes.<sup>1</sup> For comparison, traditional catalyst MoS<sub>2</sub> and 3R-MoN<sub>2</sub> containing BN impurity were also tested. For Mo-based catalysts, the surface area with the coordinately unsaturated Mo sites is catalytically active centers. Equally important is the pore structure of the catalyst, which also affects the surface area. For a given catalyst, the distribution of pore sizes may be such that the internal surface area is completely inaccessible for large reactant molecules and, which may restrict the rate of conversion to products by impeding the diffusion of reactants and products throughout the porous medium.<sup>53,54</sup> We examined the surface area and pore size for 3R-MoN<sub>2</sub> and MoS<sub>2</sub> using the BET and BJH analyses of N<sub>2</sub> physisorption



**Figure 3.** Thermal stability analysis. (a) Thermogravimetry/differential thermal analysis, TG/DTG, performed in air. Oxidation of  $\text{MoN}_x$  starts at  $\sim 610$  K, and the final product is  $\text{MoO}_3$ , which starts sublimating at a temperature of  $\sim 1000$  K. The derived nitrogen concentration is  $x = 2.2(3)$  in  $\text{MoN}_x$ . (b) TG measurement of high-temperature degassing of  $\text{N}_2$  in an Ar atmosphere. The outlet gases are qualitatively traced during heating. At  $400\text{--}445$  K, the surface absorbed  $\text{O}_2$  is released in forms of  $\text{NO}$  and  $\text{N}_2\text{O}$ . The significant degassing of  $\text{N}_2$  happens at  $\sim 850$  K.

performed at  $77$  K and low relative  $\text{N}_2$  pressure ( $P/P_0$ ) of  $0\text{--}0.3$  (see Supporting Information Figure S11).<sup>34</sup> The obtained data are listed in Table 2. For both catalysts, the measured specific surface area is  $\sim 3\text{--}10$   $\text{m}^2\cdot\text{g}_{\text{cat}}^{-1}$ ,  $\sim 10\text{--}50$  times smaller than those reported for the promoted and supported  $\text{MoS}_2$ -based catalysts.<sup>7,36,55,56</sup> Similar differences in the pore sizes are also found.<sup>55</sup> Such small values are unfavorable for achieving a high catalytic activity as already reported in well-designed catalysts.<sup>7,36,55,56</sup> Compared with  $\text{MoS}_2$  used in our comparative study,  $3\text{R-MoN}_2$  possesses a larger specific surface area, although the pore size is nearly identical (see Table 2). Based on the measured specific surface area, the catalytic activity for  $\text{MoN}_2$  can be normalized and, to a first approximation, assessed relative to the traditional  $\text{MoS}_2$ . For a more accurate assessment, we determined the active Mo sites per mole of

each of the two catalysts based on CO uptake measurement, i.e., chemisorption-desorption of CO molecules on the catalyst's surface,<sup>34,52,57</sup> and experiments were performed at similar conditions (see Experimental Section for detail). Because of the structural similarity, the two catalysts show similar desorption spectra characterized by a main peak and a weaker satellite peak (Figure 4a), corresponding to two different adsorption sites. These characteristics have theoretically been attributed to coordinately unsaturated ( $\bar{1}010$ ) and ( $10\bar{1}0$ ) Mo sites on the edge of hexagonally packed slabs.<sup>57</sup> For  $3\text{R-MoN}_2$ , the main peak occurs at a higher temperature ( $\sim 440$  K) than that of  $\text{MoS}_2$  ( $\sim 420$  K), indicating that CO molecules are more tightly bounded at unsaturated ( $\bar{1}010$ ) Mo sites. By analysis of the CO uptake data, the molar volume for the adsorbed CO can be quantified, and they are listed in Table 2 (also see Supporting Information Figure S12). The calculated ratio between  $3\text{R-MoN}_2$  and  $\text{MoS}_2$  is  $\text{CO}_{3\text{R}}:\text{CO}_{\text{MoS}_2} = 1.35:1$ , which is used to normalize the catalytic activity between two catalysts. It is worthwhile to point out that an NO uptake test was initially attempted, but the gas reacted with  $\text{MoN}_2$  to form  $\text{N}_2\text{O}$  (also see Figure 3b).

Figure 4b shows the results of HDS of DBT at  $623$  K using  $3\text{R-MoN}_2$  and  $\text{MoS}_2$ . Strikingly, at the early stages of HDS, the DBT conversion using phase-pure  $3\text{R-MoN}_2$  is more than four times higher than the conversion using  $\text{MoS}_2$ . Overall, after 2 days of the reaction,  $\sim 97\%$  DBT conversion was achieved using  $3\text{R-MoN}_2$ , whereas only  $50\%$  conversion was attained using  $\text{MoS}_2$ . The catalytic activity can more generally be described by a pseudo-first order rate constant,  $k$ , for the HDS of DBT (Figure 4c), given by

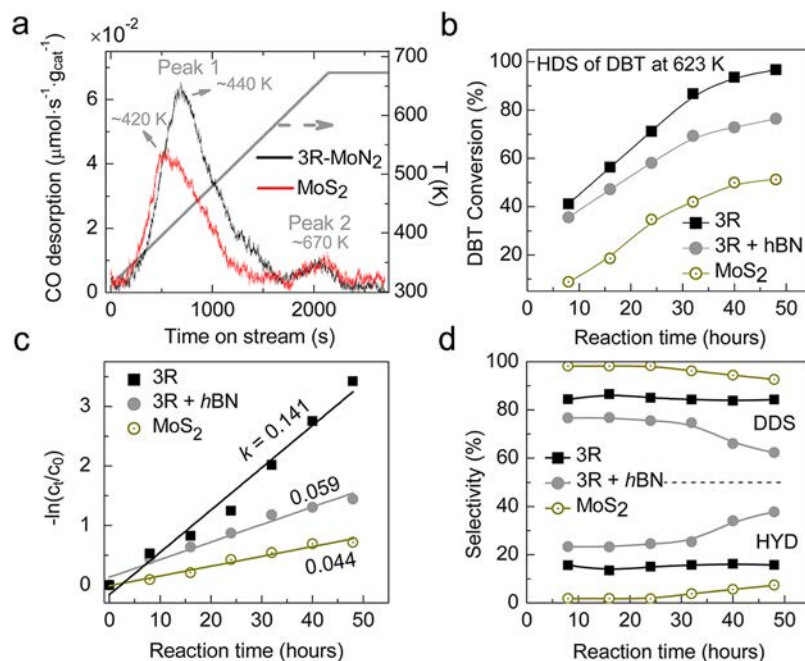
$$\ln \frac{C_t}{C_0} = -\frac{k \cdot (n_{\text{CO}})}{V_{\text{oil}}} \cdot t \quad (3)$$

where  $C_t/C_0$  denotes the ratio of DBT concentration between reaction time  $t$  and  $t = 0$ ,  $V_{\text{oil}}$  is the volume of the feed solution (i.e.,  $101.1$  mL of DBT and decalin oil), and  $n_{\text{CO}}$  is the number of active Mo sites, which can be calculated from analysis of the CO uptake data (see Table 2) and also serves as a normalization parameter the  $k$  value. The obtained value for  $\text{MoN}_2$  is  $k = 0.141$   $\text{L}_{\text{oil}} \cdot (\text{mol}_{\text{CO}})^{-1} \cdot \text{s}^{-1}$ , which is more than three times higher than that of  $\text{MoS}_2$  ( $\sim 0.044$   $\text{L}_{\text{oil}} \cdot (\text{mol}_{\text{CO}})^{-1} \cdot \text{s}^{-1}$ ). Therefore,  $3\text{R-MoN}_2$  is substantially more catalytically active than the traditional  $\text{MoS}_2$  catalyst, presumably due to unsaturated Mo atoms that are more energetically enhanced at ( $\bar{1}010$ ) sites in  $3\text{R-MoN}_2$  (see Figure 4a). It is worth mentioning that, to determine the intrinsic catalytic activity, neither promoter nor support was used for  $3\text{R-MoN}_2$  and  $\text{MoS}_2$ . However, the catalytic activity of  $3\text{R-MoN}_2$  is significantly deteriorated in the presence of  $h\text{BN}$  impurity

**Table 2.** Summary of CO Uptake Data, BET Surface Area, and Pore Volume and Width for  $3\text{R-MoN}_2$  and  $\text{MoS}_2$  Catalysts<sup>a</sup>

catalyst	$3\text{R-MoN}_2$	$\text{MoS}_2$	ratio $3\text{R}:\text{MoS}_2$
BET surface area [ $\text{m}^2\cdot\text{g}_{\text{cat}}^{-1}$ ]	8.656 (9)	3.699 (5)	2.34
pore volume [ $\mu\text{m}^3\cdot\text{g}_{\text{cat}}^{-1}$ ]	0.036	0.016	2.25
Average pore width [nm]	16.54	17.14	0.96
CO uptake [ $\mu\text{mol}_{\text{CO}}\cdot\text{g}_{\text{cat}}^{-1}$ ]	47.19	34.97	1.35
CO/Mo [ $\times 10^{-3}$ $\text{mol}_{\text{CO}}\cdot\text{mol}_{\text{Mo}}^{-1}$ ] <sup>b</sup>	5.850	5.598	1.05
$k$ [ $\text{L}_{\text{oil}}\cdot\text{mol}_{\text{CO}}^{-1}\cdot\text{s}^{-1}$ ]	0.141	0.044	3.20

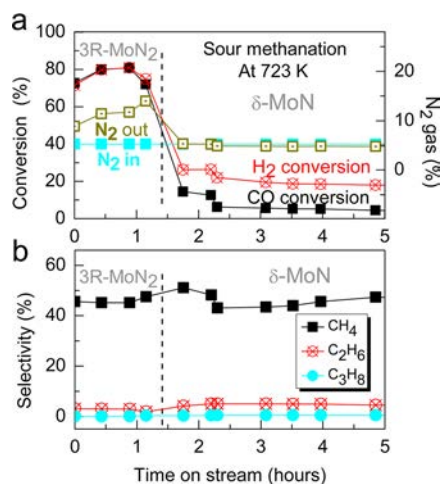
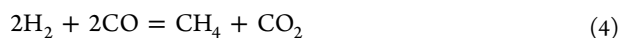
<sup>a</sup>Also listed are the pseudo-first-order rate constant,  $k$ , which are normalized to the active Mo sites. <sup>b</sup>CO in the CO/Mo ratio denotes the molar volume of CO adsorbed onto the Mo sites (i.e., coordinately unsaturated Mo sites). The ratio CO/Mo represents the percentage of catalytically active Mo sites.



**Figure 4.** Catalytic activities of 3R-MoN<sub>2</sub> in the hydrodesulfurization (HDS) of dibenzothiophene (DBT). Traditional catalyst MoS<sub>2</sub> was also tested for comparison. (a) Temperature-programmed desorption (TPD) of CO to determine the amounts of active Mo sites. Before experiment, the catalysts were pre-exposed to CO gas at 323 K for 60 min to absorb CO molecules and form Mo-CO bonds with active Mo sites, including coordinately unsaturated ( $\bar{1}0\bar{1}0$ ) and ( $10\bar{1}0$ ) Mo sites. (b) HDS of DBT at 623 K and 6 MPa in H<sub>2</sub> atmosphere. (c) Relative DBT concentration,  $c_t/c_0$ , as a function of reaction time  $t$ . The normalized pseudo-first-order rate constant,  $k$ , was deduced by linear fits. (d) The measured selectivity of direct desulfurization (DDS) and hydrogenation (HYD) in the HDS of DBT (see Supporting Information Figures S9–S14).

(see XRD in Supporting Information Figure S9), indicating that nonmetallic impurities would lead to the reduction of active sites for the catalytic reaction. On the other hand, as shown in Figure 4d, the BN-MoN<sub>2</sub> catalyst exhibits an enhanced catalytic activity for the hydrogenation (HYD) of DBT to form sulfur-free products of cyclohexylbenzene (CHB) and bicyclohexyl (BCH) (see Supporting Information Figure S10). For direct desulfurization (DDS), biphenyl (BP) is the only product. Compared with MoS<sub>2</sub>, the BN-MoN<sub>2</sub> catalyst is 6–12 times more selectivity to HYD. Besides, the phase-pure 3R-MoN<sub>2</sub> exhibits selectivity to HYD that is 2–7 times higher than MoS<sub>2</sub>, close to the case of the promoted and supported Ni-MoS<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub>, a widely used commercial catalyst.<sup>58</sup> For both BN-MoN<sub>2</sub> and phase-pure MoN<sub>2</sub>, the reported ranges of selectivity reflect their sensitivity to reaction time (Figure 4d), which may be associated with the DBT concentrations, as the latter decrease with increasing reaction time. Clearly, 3R-MoN<sub>2</sub> is an intrinsic supercatalyst for the hydrogenation, which is highly attractive from the perspectives of ultra-deep hydrodesulfurization of fossil fuels.

The catalytic activities were also evaluated using MoN<sub>2</sub> as a catalyst in the sour methanation of syngas, a synthesis gas consisting primarily of CO, H<sub>2</sub>, and sour gases such as H<sub>2</sub>S, which often causes catalyst poisoning when traditional Ni-based catalysts were used.<sup>2</sup> Our preliminary results of sour methanation of syngas are shown in Figure 5a. At the early stages of the reaction, more than 80% conversion was achieved for both H<sub>2</sub> and CO at 723 K in a H<sub>2</sub>S atmosphere. As the reaction proceeds, the outlet gases of CH<sub>4</sub> and CO<sub>2</sub> were detected in a molar ratio CH<sub>4</sub>:CO<sub>2</sub> ≈ 1:1, indicating a sour methanation process given by



**Figure 5.** Catalytic performance of 3R-MoN<sub>2</sub> in sour methanation. (a) Conversion of H<sub>2</sub> and CO as a function of reaction time at 723 K. Volume fractions of the input and output N<sub>2</sub> gas are plotted to show the high- $T$  degassing of MoN<sub>2</sub> and the subsequent formation of  $\delta$ -MoN. (b) The measured selectivity of CH<sub>4</sub>, C<sub>2</sub>H<sub>6</sub>, and C<sub>3</sub>H<sub>8</sub> in the sour methanation. No promoters or supporting substrates were used in the catalytic measurements.

The observed high catalytic activity of 3R-MoN<sub>2</sub> for sour methanation is comparable to that obtained recently using a Zr-promoted MoS<sub>2</sub> catalyst<sup>2</sup> and is significantly higher than the catalytic activities of Co-MoS<sub>2</sub>/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>.<sup>37</sup> However, the reaction 4 lasted for only 1.4 h, after which the conversion for both H<sub>2</sub> and CO was lowered drastically. Concurrently, as shown in Figure 5a, the excessive N<sub>2</sub> was released as the reaction proceeds. Most likely, this is due to the degassing of 3R-MoN<sub>2</sub> at 723 K, which converts 3R-MoN<sub>2</sub> to hexagonal



$\delta$ -MoN. As the  $N_2$  output is stabilized at the input level,  $H_2$  and CO are converted into  $CH_4$  in a molar ratio of  $H_2:CO \approx 3:1$ , which is typical of a traditional methanation reaction



Obviously, 3R-MoN<sub>2</sub> is catalytically more active and efficient than  $\delta$ -MoN in the methanation of syngas. The 3R-MoN<sub>2</sub> and  $\delta$ -MoN catalysts, however, show similar selectivity for  $CH_4$  as seen in Figure 5b. In practice, the syngas can be produced by gasifying coal and biomass. The sour methanation is an essential process for upgrading such low-grade energy sources. The search for new catalysts with high poisoning resistance is highly demanded from the perspectives of enhanced process and energy efficiency. As a candidate catalyst, the 3R-MoN<sub>2</sub> holds great promise for future sour methanation applications.

To the best of our knowledge, 3R-MoN<sub>2</sub> is the first binary nitride having a rhombohedral MoS<sub>2</sub> structure and is only the fifth binary nitride in bulk forms exhibiting the stoichiometry of  $XN_2$ , after noble metal nitrides ReN<sub>2</sub>, OsN<sub>2</sub>, IrN<sub>2</sub>, and PtN<sub>2</sub>.<sup>25,27,59</sup> As expected, the layer-structured 3R-MoN<sub>2</sub> is intrinsically an excellent catalyst. It is sulfur-poisoning tolerant and exhibits superior catalytic activities over the traditional MoS<sub>2</sub>-based catalysts for the HDS of DBT, even without the presence of a promoter or a supporting substrate. It is conceivable that the discovery of 3R-MoN<sub>2</sub> may open new opportunities for developments of highly active, MoN<sub>2</sub>-based catalysts through the introduction of promoter species (e.g., Co, Ni, and Zr) and supporting substrates (e.g., Al<sub>2</sub>O<sub>3</sub> and SiO<sub>2</sub>), as already demonstrated for the MoS<sub>2</sub> catalysts.<sup>1,2</sup> As another nitride of the group 6 metals, WN<sub>2</sub> may be structurally isotypic with 3R-MoN<sub>2</sub> and, hence, a promising catalyst candidate, although an R3 structure has previously been reported.<sup>33</sup> The crystal structure of WN<sub>2</sub>, however, has not been well resolved because the sample used in ref 33, prepared by evaporating W filaments in atmospheres of N<sub>2</sub>, is poorly crystalline. More studies are required to resolve the structure of WN<sub>2</sub>. The ternary nitrides of LiMoN<sub>2</sub> and LiWN<sub>2</sub> can be another group of materials with good catalytic activities. They also crystallize in a R3m structure, which can simply be viewed as the intercalation of Li ions into the interlayer of  $XN_2$  to form octahedrally coordinated [LiN<sub>6</sub>].<sup>42,60</sup> These ternary nitrides are thermodynamically more favorable than binary 3R-MoN<sub>2</sub>. For example, at atmospheric pressure, the degassing temperature of LiMoN<sub>2</sub> ( $\sim 1023$  K)<sup>41</sup> is  $\sim 20\%$  higher than that of 3R-MoN<sub>2</sub> ( $\sim 850$  K). This high thermal stability may make them robust catalyst in high-temperature catalytic application.

## CONCLUSIONS

In summary, we have successfully synthesized a novel molybdenum nitride compound, 3R-MoN<sub>2</sub>, which has the rhombohedral MoS<sub>2</sub> structure, through a high  $P$ - $T$  route of solid-state ion-exchange. 3R-MoN<sub>2</sub> adopts a peculiar  $4d^{2.5}$  electron configuration with a non-integral valence of +3.5 for Mo ions. Most importantly, 3R-MoN<sub>2</sub> exhibits superior catalytic activities and high hydrogenation selectivity over MoS<sub>2</sub> in the hydrodesulfurization of dibenzothiophene. Further, the hydrogenation selectivity of this nitride can be greatly improved when the  $h$ BN impurity is present. 3R-MoN<sub>2</sub> is also catalytically more active and efficient than  $\delta$ -MoN in the sour methanation of syngas. Hence, this compound holds great promise as the next-generation catalyst for a wide range of applications. The new nitride can be synthesized at a

moderately high pressure of 3.5 GPa, making it feasible for massive and industrial-scale production.

## ASSOCIATED CONTENT

### Supporting Information

Detailed sample characterizations (including XRD, SEM, EDX, NPD, and XPS measurements) and measurements on phase stability, density, and catalytic properties. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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### Notes

The authors declare no competing financial interest.

## ACKNOWLEDGMENTS

This work was partially supported by UNLV High Pressure Science and Engineering Center (HiPSEC), which is a DOE NNSA Center of Excellence operated under Cooperative Agreement DE-FC52-06NA27684, and UNLV start-up funding to Y.Z. The work was also partially supported by the NSF of China (No. 21473231). Initial work was supported by Los Alamos National Laboratory, which is operated by Los Alamos National Security LLC under DOE Contract DE-AC52-6NA25396. Use of the ORNL's Spallation Neutron Source at B1 beamline (NOMAD) was sponsored by the Scientific User Facilities Division, DOE-BES. The work was also partially supported by the Strategic Priority Research Program of the CAS for clean and efficient utilization of low-rank Coal, under Grant No. XDA07020400.

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